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Release of coumarin encapsulated in chitosan-gelatin irradiated films

19 **Abstract**

20 Chitosan and fish gelatin, by-products from marine industry, were used to prepare active 21 biobased films containing an antioxidant (coumarin). After drying, the films were irradiated at 40 22 and 60kGy by electron beam accelerator. The effect of irradiation on the film properties as well 23 as the antioxidant release mechanism were investigated and compared with the control. Electron 24 Spin Resonance (ESR) unravelled free radical formation during irradiation in films containing 25 coumarin. After antioxidant addition and/or irradiation treatment, only a shift of amide A, and 26 amide B peak was observed for all the films, and a shift of amide II band for the control film 27 after 60kGy irradiation dose. Irradiation allowed to improve the thermal stability of the control 28 films. Both addition of coumarin and irradiation increased the surface wettability (increase of the 29 polar component of the surface tension). From the water barrier analysis, neither irradiation nor 30 coumarin addition influenced the permeability at the lower RH gradient used (0-30% RH). Using 31 the higher RH gradient (30-84%) induced a rise of the WVP of all films (containing or not 32 coumarin) after irradiation treatment. At 60kGy, the tensile strength of only the control films 33 increased significantly. Finally, even if functional and structural properties are only weakly 34 affected, it is enough to modify the release kinetics of the antioxidant into aqueous medium. The 35 apparent diffusion coefficient of coumarin is two times reduced after irradiation. Irradiation also 36 allowed to better protecting the encapsulated antioxidant. Indeed, the amount of coumarin in the 37 non-irradiated film was significantly lowered Compared to the initial quantity, which is probably 38 due to degradation. Coumarin in irradiated films is more protected considering this aspect.

39

40 **Key Words:** Electron beam irradiation; coumarin; chitosan-fish gelatin edible film; functional 41 and structural properties, release properties

- 42
- 43

44 **1. Introduction**

45 Maintaining food quality, improving safety, and reducing storage losses waste are key objectives 46 of a sustainable food system. Nowadays, the modern food industry is facing new challenges one 47 of which being related to the food packaging to extend shelf life. Currently, a great number of 48 research works are focused on the use of bio-based films with good water and oxygen barrier 49 properties to protect food (Fabra et al., 2011). Since the consumer demand has shifted to safe 50 materials, especially from renewable agriculture by-products and food processing industry 51 wastes (Tharanathan, 2003), natural polymers (proteins, polysaccharides) are the main 52 components of edible films (Gontard & Guilbert, 1994). They are considered as active packaging 53 when they incorporate active compounds, such as antimicrobials, preservatives or antioxidants, 54 which allow to improve food quality and safety (Han, 2002). Chitosan is a natural polymer from 55 fish industry waste obtained by the deacetylation of chitin. It is a nontoxic material, 56 biocompatible, and biodegradable that manifests antibacterial properties. In acidic environment 57 the amino groups are protonated and their positive charges can interact with polyanions such as 58 alginate, carrageenan, gelatin, etc. forming polyelectrolyte complexes (PEC) increasingly used in 59 the encapsulation of various biocomponents. Due to these characteristics, chitosan has been 60 widely used for the production of edible films as well as bio-compatible polymeric materials 61 (Aider, 2010; Rivero et al., 2010). Gelatin is another widely used bio-based material obtained by 62 the controlled hydrolysis of the insoluble fibrous collagen present in the bones and skin 63 generated as waste during animal slaughtering and fish processing. Its excellent film forming 64 ability is well-known (Hoque et al, 2011). Gelatin-based films are used for coating or packaging 65 in order to maintain the quality of foods during storage, due to its good barrier to oxygen, light 66 and prevention of dehydration and lipid oxidation (Jongjareonrak et al, 2006). Most research on 67 gelatin film has focused on gelatin derived from mammalian sources such as bovine and porcine. 68 Recently, there has been more interest in using fish as alternative sources of gelatin, due to 69 religious considerations or fear of bovine spongiform encephalopathy (Pérez-Mateos et al.,

70 2009).

71 Chitosan and gelatin have been shown to be compatible due to the ability to associate through 72 electrostatic and hydrogen bonding. Specifically, when chitosan is positively charged and gelatin 73 is negatively charged under appropriate conditions of pH. This is particularly important to 74 improve the final network properties as compared to those obtained from the pure polymers 75 (Benbettaieb et al., 2015a). Thus, many investigations focused on their possible use as a matrix 76 to obtain bio-packaging materials. In this sense and to make films even more useful, functional 77 edible films that contain active compounds have been developed, to enhance food quality and 78 product shelf-life (Suppakul et al, 2003). The incorporation of antioxidants in these 79 biodegradable and edible polymers is an interesting alternative to food preservation, since 80 oxidation is one of the major problems affecting food quality as well as film biopolymer stability 81 during ageing (Martins et al, 2012). The use of natural, non-toxic antioxidants such as ferulic 82 acid or α -tocopherol to preserve the consumer health has been investigated $(Fabra et al, 2011)$. 83 Several researchers have previously reported on the potential benefits of using naturals 84 antimicrobials and antioxidants compounds in edible and bio-based films for extending food 85 shelf life (Oussalah et al., 2004; Ouattara et al., 2000). However, little information exists about 86 the influence of these compounds on films structural and physicochemical properties.

87 Recently, Tammineni et al.(2014) reported that mechanical and barrier properties of bovine 88 gelatin films were improved after tannic acid incorporation. Furthermore, crosslinking bovine 89 gelatin films with tannic acid results in reduction of film solubility by about 80% (Zhang et al., 90 2010). Kavoosi et al. (2014) studied antioxidant and antibacterial activity of gelatin films 91 incorporated with *Zataria multiflora* essential oil (2 to 8% w/w of gelatin). They reported that 92 beside their excellent antibacterial properties against both Gram-positive and Gram-negative 93 bacteria, bioactive films have new functional properties. Peng and Li. (2014) demonstrated that

94 water vapour permeability of chitosan films decrease while the tensile strength inversely 95 increases when essentials oils are incorporated. Therefore, we were interested in encapsulating 96 these natural compounds into chitosan-gelatin blend edible films. Moreover, physical methods 97 including dehydrothermal treatment, ultraviolet, heat and gamma irradiation (Bigi et al., 1998) 98 help to modify the polymeric network through the cross-linking of the polymer chains and also 99 help to improve the functionality of polysaccharide (Sabato et al., 2000) or protein (Vachon et 100 al., 2000) based films. Indeed, irradiation treatments have been described to enhance water 101 barrier and mechanical properties of protein-polysaccharides complexes (Lacroix et al., 2002; 102 Lee et al., 2004; Jo et al., 2005; Inamura et al., 2013). The structural modifications induced by 103 irradiation could increase the capacity of cross-linked edible films to control the release of 104 embedded active compounds. In the current literature there is a lack of detailed studies dealing 105 with the effects of polymers structure, in particular chitosan-gelatin films, on the retention and 106 release properties of the added antioxidant compounds (Papadokostaki et al, 1997). Very few 107 studies have been published on the impact of irradiation on the release of active compounds from 108 natural biopolymers. Indeed, Tin Wui et al. (2002) showed that the release-retarding property of 109 alginate and alginate–chitosan beads is significantly enhanced after the beads irradiated by 110 microwave. In the same way, Lacroix et al. (2002) displayed that gamma-irradiation induces 111 cross-links in calcium caseinate edible films and thus allows a better control of enzyme and 112 active compounds release. Previous works displayed that ferulic acid, quercetin or tyrosol 113 addition affected differently the functional properties of gelatin-chitosan films according the 114 irradiation dose (Benbettaieb et al, 2015b).Irradiation accentuated the wettability and the 115 hydrophilicity of the film containing antioxidants whereas oxygen barrier and thermal stability 116 were enhanced.

117 The aim of this study is to further investigate the effect of coumarin addition and electron beam 118 irradiation on the mechanical, thermal, barrier and structural properties of chitosan-fish gelatin 119 edible films. The effect of irradiation on the coumarin release in liquid medium was also 120 investigated.

121

122 **2. Materials and methods**

123 *2.1. Materials and reagents*

124 Commercial grade chitosan (CS) (France Chitine, , MW=165 kDa, low viscosity, 85%, 125 deacetylation degree, France) and commercial grade fish gelatin (G) (Rousselot 200 FG 8 with a 126 180 Bloom degree, a viscosity of 4 mPa.s at 45°C and with the concentration of 6.67% ,in water, 127 and a pH=5.4) were used as film-forming matrix. Anhydrous glycerol (GLY) (Fluka Chemical, 128 98% purity, Germany) was used as a plasticizer in order to improve the mechanical properties of 129 the films. Glacial acetic acid (Sigma, 99.85% purity) was used to prepare the solvent for chitosan 130 and helped to improve their solubility. Silica gel and potassium chloride saturated salt solution 131 (KCl, Sigma-Aldrich, France) were used to fix the relative humidity at <2% and 84% for water 132 vapour permeability measurements. Coumarin (minimum purity 99%, Sigma Aldrich, molecular 133 weight = 146 g.mol⁻¹, molar volume = 117 cm³.mol⁻¹, melting point = 70°C, LogP = 1.39, 134 solubility in water = 1.9 g.L⁻¹ (at 25°C), boiling point =298°C (à 101325 Pa), surface tension= 135 46.4 mN.m⁻¹ (at 25°C), saturated vapour pressure = 1.33 Pa (at 25°C), data from 136 Chemspider.com, was used as a model of natural antioxidant molecule.

137

138 *2.2. Film formation*

139 20 grams of the chitosan powder was dispersed in 1 L of a 1% (v/v) aqueous acetic acid to obtain 140 a 2% (w/v) film forming solution. The solution was homogenized at 1200 rpm with high shear 141 homogeniser (Ultra Turrax (RW16 basic- IKA-WERKE) at 25°C. As a clear film-forming 142 solution was obtained, no more treatment has been applied to the chitosan solution for improving 143 the solubilisation. Then, 2.2 grams of glycerol (10% w/w dry matter) was added to this solution,

144 under stirring. The pH of the chitosan solution was about 4.9±0.2. A 60 grams of fish gelatin 145 powder was separately solubilized in 1 L of distilled water under continuous stirring and heating 146 at 70°C for 30 min to obtain a 6% w/v solution (pH≈6.5). 6.6 grams of Glycerol (10% w/ dry 147 matter) was added to this film forming solution after complete solubilisation of gelatin.

148 Subsequently, equal weight of the respective solution was mixed at 1:1 ratio and stirred for 30 149 min and pH was adjusted to 5.6 with acetic. This condition was specifically designed to obtain a 150 polyelectrolyte complex between chitosan and gelatin since the iso-electric point of gelatin (Ip= 151 4.5-5.2) while the pKa of the amino group (pH=6.2-6.5) of chitosan. At this condition, therefore, 152 gelatin is negatively charged while chitosan is positively charged thus avoiding any phase 153 separation upon mixing. Coumarin was added to the final film forming solution at a 154 concentration of about 50 mg /g polymer (corresponding to a 47 mg/g total dry weight of film). 155 The aqueous dispersions were homogenized at 1200 rpm using the Ultra Turrax until complete 156 dissolution.

157 30 mL of the film forming solution (FFS) in the presence and absence of coumarin was then 158 poured into plastic Petri dishes (13.5 cm diameter). A minimum of 30 films (ie 30 Petri dishes) 159 have been prepared for each formulation. Aqueous solvent was removed by drying in a 160 ventilated climatic chamber (KBF 240 Binder, ODIL, France) at 25°C and 45% RH for 18 to 161 24h. After drying, films were peeled off from the surface and stored up to equilibrium in a 162 ventilated climatic chamber (KBF 240 Binder, ODIL, France) at 50% RH and 25°C before each 163 measurements.

164

165 *2.3. Radiation treatment*

166 Irradiation was carried out in the AERIAL pilot plant (Innovation Park, Illkirch, Strasbourg, 167 France), using a linear electron accelerator at ambient temperature (20±0.5ºC). Thin dried films 168 (70 to 85µm thickness) were irradiated for 40 and 60kGy using a 2.2 MeV energy electron beam 169 at a dose rate of 0.3 kGy/sec. One batch of film formation was preserved as non-irradiated 170 reference. We selected a maximum of 60kGy doses that can yield a high density of crosslinking 171 in proteins and in the range of authorized doses in food and packaging. Moreover, according to 172 the Codex General Standard for Irradiated Foods (CAC, 2003), ionizing radiations foreseen for 173 food processing are recommended to be limited to 60kGy. Dosimetry was performed using 174 alanine pellet dosimeters calibrated according to ASTM/ ISO 51607 (2004).

175

176 *2.4. Film characterizations*

177 *2.4.1. Thickness measurement*

178 Film thickness was measured with an electronic gauge (PosiTector 6000, DeFelsko Corporation, 179 USA). Five measurements were taken for each film sample, one from the center and four close to 180 the perimeter. Mean value was used in further calculations. The film thickness according the 181 sample and formulation ranges between 70 and 85

182 *2.4.2. Mechanical properties*

183 A universal traction testing machine (TA.HD plus model, Stable MicroSystems, Haslemere, 184 England) with a 300 N load cell was used to determine tensile strength (TS, MPa), Young's 185 modulus (YM, MPa) and percentage of elongation at breakpoint (E, %), according to ASTM 186 method D882 (1992). Rectangular film samples $(2.5 \times 8 \text{ cm}^2)$ were cut using a special precision 187 sample cutter (Thwing-Albert JDC Precision Sample Cutter) in order to get tensile test piece 188 with an accurate width and parallel sides throughout the entire length. Before testing, all samples 189 were equilibrated for two weeks at 25°C and 50% relative humidity (RH). Equilibrated film 190 samples were then placed in the extension grips of the testing machine and stretched uniaxially at 191 a rate of 50 mm/min until breaking. TS, YM and E were determined from stress–strain curves. 192 Measurements were carried out at room temperature $(25\pm2\degree C)$ and five samples for each 193 formulation were tested.

194 *2.4.3. Water vapour permeability*

195 The water vapour permeability (WVP) was determined according to the gravimetric method 196 described in the ASTM E96-80 (1980) and adapted to edible and bio-based films by Debeaufort, 197 Martin-Polo and Voilley (1993). Two relative humidity gradients were used: 0-30% and 30-84%. 198 Prior to WVP measurements, all film samples were equilibrated at $25 \pm 0.5^{\circ}$ C and 30% relative 199 humidity for 72 h. The average value of five thickness measurements per type of film was used 200 in the WVP calculations (statistical error on the film thickness was taken into account in WVP 201 uncertainty). The film samples (6.44 cm² discs) were placed between two teflon rings on the top 202 of the glass cell containing silica gel (0% RH) for the first RH gradient (0-30%) or a salt solution 203 of KCl (84% RH) for the second RH gradient (30-84%). The second RH gradient was selected to 204 obtain an average water content of film at the stationary state of permeation that corresponds to 205 the RH of 50% used for mechanical property characterizations and FTIR analysis. Those 206 permeation cells were then introduced into a climatic chamber (KBF 240 Binder, ODIL, France) 207 maintained at 30% RH and $25\pm0.5^{\circ}$ C and periodically weighed.

208 The WVP $(g.m^{-1}.s^{-1}.Pa^{-1})$ calculation was based on the change in the absolute value in weight 209 loss of the permeation cell versus time once the steady state was reached (Benbettaieb et al, 210 2015a). Five replicates for each film formulation were performed.

211 *2.4.4. Surface properties*

212 *2.4.4.1. Water contact angle*

213 The sessile drop method was used for contact angle measurement, with a DGD-DX goniometer 214 (GBX, Romans-sur-Isere, France), equipped with the DIGIDROP image analysis software 215 (GBX, Romans-sur-Isere, France), according to Karbowiak et al. (2006). A droplet of liquid (\sim 216 1.5 µL) was deposited on the film surface with a precision syringe. Then, the method is based on 217 image processing and curve fitting for contact angle measurement from a theoretical meridian 218 drop profile, measuring contact angle between the baseline of the drop and the tangent at the 219 drop boundary. The contact angles with water and diiodomethane at 0 and 20 s were measured 220 from both sides of the drop and averaged. The measurement was carried on over 120 s. Five 221 replicates per film were carried out.

222 *2.4.4.2. Surface tension*

234

223 The surface tension (γ_S) of the film and its polar (γ_S^p) and dispersive (γ_S^d) components were 224 calculated using diiodomethane (γ_L = 53 mN/m; γ^d_L = 50.8 mN/m and γ^p_L = 0 mN/m) and 225 water $(\gamma_L = 72 \text{ mN/m } ; \gamma_L^d = 21.8 \text{ mN/m and } \gamma_L^p = 51 \text{ mN/m}$ knowing their surface 226 tension (γ_L) and respective dispersive (γ_L^d) and polar components (γ_L^p) given by Strom et 227 al.(1987) and according the following equations established by Owens and Wendt (1969) :

γ γ- γ- 228 ………………………………………………… Eq.1

229
$$
\gamma_{Li}(1 + cos\theta i) = 2\left[\sqrt{\gamma_S^d \times \gamma_{Li}^d} + \sqrt{\gamma_S^p \times \gamma_{Li}^p}\right] \dots \dots \dots \text{Eq.2}
$$

230 where the subscripts S and L refers to the solid (the film surface) and the liquid, respectively. As 231 only two liquids have been used for the regression, the accuracy of values obtained from this 232 analysis has been considered at the p-level of 0.01.

233 *2.4.4.3. Swelling index and swelling rate with water*

235 Swelling index was obtained from the water drop volume kinetics using the following equation:

236 Swelling index (%) =
$$
\left(\frac{\Delta V}{V_0}\right)
$$
. 100 = $\left(\frac{V_{\text{max}} - V_0}{V_0}\right)$. 100 Eq.3

237 where ΔV is the droplet volume variation (μ L) measured on the film sample (over the first 20s).

238 V_{max} is the maximal apparent volume (μ L) of the droplet and V₀ is the initial volume (μ L) of the 239 droplet.

240 Swelling rate was obtained from the drop volume kinetics using the following equation:

241 Swelling rate
$$
(\mu L/s) = \left(\frac{\Delta V}{\Delta t}\right) = \frac{V_{\text{max}} - V_{\text{ini Sw}}}{t_{\text{max}} - t_{\text{ini Sw}}}
$$
 Eq.4

- 242 Where ∆V is the droplet volume variation (µL) during ∆t time (s) measured on the film sample,
- 243 assuming a linear regression.
- 244 V_{max}: maximum swelling volume, V_{ini Sw}: initial volume of swelling, t_{max}: time of maximum 245 swelling, $t_{\text{ini Sw}}$: initial time corresponding to the beginning of swelling
- 246

247 *2.4.5. GPC-MALLS system*

248 The GPC-MALLS system consisted of a degasser ERC-3215- α (ERC, Japan), a constametric \circledR 249 3200 MS pump (Thermo Separation Products, FL), an injection valve with 100 µL loop 250 (Reodyne 7725i) fitted inside a temperature regulated oven (Gilson, Model 831, UK) maintained 251 at 40° C $\pm 1^{\circ}$ C and a DAWN-DSP multi-angle light scattering photometer (Wyatt Technology, 252 Santa Barbara, CA, USA) equipped with He–Ne laser $(\lambda = 633 \text{ nm})$. Simultaneous concentration 253 detection was performed using a calibrated differential refractometer (RI 2000, Schambek, 254 Germany). A refractive index increment dn/dc value of 0.180 mL/gm was used in the 255 calculations

256 The mobile phase was made to contain 0.15M ammonium acetate, 0.2M acetic acid and 0.1M 257 sodium chloride and was filtered through 0.2 µm pore size cellulose nitrate membrane. The 258 samples injected were subjected to prior filtration through a nylon filter of 0.45 µm pore size. A 259 set of two columns SB-803HQ and SB-806HQ (8 mm × 300 mm, Shodex OHpak, Japan, 260 exclusion limits 1×10^5 and 2×10^7 g/mol), housed inside the oven, was used for the separation. 261 The flow rate for the eluent was 0.45 mL/min. The Berry fitting method with linear fit was used 262 for data processing in ASTRA software (Version 4.90.08). For the measurement of molecular 263 weight, about 27mg (30mg of films containing about 12% water content) of chitosan-gelatin film 264 incorporated with coumarin 5% (w/w polymer) was dissolved in 10mL of mobile phase. The 265 solution was then heated in a water bath for 20 min at 45° C and subsequently centrifuged for 266 5min at 25000 rpm at 25°C. The soluble fraction was removed from the respective solution and

267 was heated at 45°C for 5 min prior to injection into the GPC-MALLS system.

268 The entire GPC-MALLS system was maintained at 40 ± 1 °C. Temperature control was achieved 269 using the in-house heating methodology provided thermostatic heating to all pipework between 270 the detectors.

271 *2.4.6. Thermogravimetric analysis*

272 Thermogravimetric analysis (TGA) was used to evaluate the thermal stability of the samples. 273 Measurements were performed using a TA instrument (TA instruments Discovery TGA New-274 Castle, USA), from 25 to 800°C, at a heating rate of 20°C/min, under nitrogen atmosphere. The 275 weight of the film sample (initially around 8 mg) was constantly measured with an accuracy of 276 0.01 mg. Films were stored at 25°C and 50% RH for two weeks before TGA measurements.

277 *2.4.7. Attenuated Total Reflectance - Fourier Transforms Infrared (FTIR-ATR) spectroscopy*

278 The Fourier Transform Infrared spectra from each film were obtained using a spectrometer 279 (Perkin-Elmer, Spectrum 65, France) using Attenuated Total Reflectance (ATR) with ZnSe 280 crystal. For each measurement, 32 scans in the wave length range 400-4000 cm⁻¹ with a 4 cm⁻¹ 281 resolution were co-added before the Fourier transform. The spectra were collected in duplicate. 282 This analysis aimed at determining the modifications at the molecular scale induced between 283 active molecules and polymers after electron beam irradiation.

284 *2.4.8. Electron spin resonance (ESR)*

285 The ESR technique allows measuring the presence of free radicals. ESR signals were recorded at 286 room temperature on a Bruker EMX spectrometer (Bruker, Berlin, Germany) controlled with a 287 Bruker ER 041 XG microwave bridge operating at X-band (~9 GHz). 50mg of each film was cut 288 and placed in the sampling tube. ESR spectra were carried out using 100 kHz magnetic field with 289 6G modulation amplitude, 20.12 mW microwave power, 9.49 GHz frequency and 5 scans.

290 *2.5. Release of coumarin in aqueous medium*

291 Prior to release experiments, the real concentration of coumarin in the films was determined. A 292 film sample of about 60 mg (about 5x5 cm²) was immersed up to fully dispersion in 100 mL of 293 acetic acid solution at pH 4 under stirring at 50° C in order to fully solubilized the polymers and 294 the coumarin. The amount of coumarin in the liquid medium was determined by UV-vis 295 spectrophotometry (Biochrom Libra S22) at 278 nm (previously determined from the absorbance 296 spectrum of the pure coumarin in water). A series of standard solutions for this antioxidant (1, 2, 297 4, 5, 10, 25 and 50 mg/L) was used for calibration, according to the Beer-Lambert's law. This 298 concentration was then compared to the theoretical content introduced in the films when 299 prepared. The measured initial concentration was used to calculate the percentage of retention in 300 the film after the release kinetics.

301 The release of the coumarin was carried out in triplicate using the rotating paddle dissolution 302 apparatus (AT7 Smart type II, Sotax, Basel). 600 mg of each film were incorporated at time 0 in 303 1 L of the dissolution medium (water adjusted at pH=7 using 0.1 M of NaOH). 3 mL samples 304 were withdrawn and assayed for antioxidant release periodically up to equilibrium. The amount 305 of coumarin in the release medium was determined by UV-vis spectrophotometry (Biochrom 306 Libra S22) as previously described.

307 The effective diffusion coefficient of coumarin in the film (D) was also determined from the 308 release kinetics assuming a Fick's law (Eq.5), and considering the transient state of the transfer.

$$
\frac{\partial C}{\partial t} = D \frac{\partial^2 C}{\partial x^2}
$$
 Eq.5

310 Where C is the concentration of this antioxidant in the film over the time t, x is the thickness of 311 the film and D is the effective diffusion coefficient.

312 The experimental method chosen corresponds to the case of diffusion of the solute from a plane 313 sheet (film) into a stirred solution of limited volume (Crank, 1975). As the solution is constantly 314 stirred, we assumed there is no boundary layer effect on diffusion. Therefore, the concentration 315 of coumarin in the solution, initially zero, is considered to be uniform in the release medium

316 according to Crank (1975). The concentration of this antioxidant in the film is also assumed to be 317 uniformly distributed within the film at time zero. We also consider a unidirectional diffusion of 318 the coumarin in the film, and a D which does not depend on the concentration or on the time. 319 This mass transfer equation (Eq.5) can thus be solved using the following analytic solution to the 320 second Fick's law applied to transient state (Crank, 1975):

$$
\frac{C_t}{C_{\infty}} = 1 - \sum_{n=1}^{\infty} \frac{2\alpha(1+\alpha)}{1+\alpha+\alpha^2 q_n^2} \exp\left(-\frac{Dq_n^2 t}{l^2}\right)
$$
 Eq. 6

321 Where, Ct is the concentration of coumarin determined in the dissolution medium over time, as 322 previously detailed; C∞ is the maximum concentration of this antioxidant determined in the 323 dissolution medium when equilibrium is achieved; $\alpha = V_s/(K \times V_f)$ with V_s the volume of 324 solution (m³), V_f the volume of the film (m³) and K the partition factor; q_n are the non-zero 325 positive roots of $tan(q_n) = -\alpha q_n$ using n values between 1 and 6.

326 D is the effective diffusion coefficient $(m^2.s^{-1})$, and l is the half thickness of the film (m).

327 This model was applied to the release experimental kinetics (up to equilibrium) in order to 328 determine the effective diffusion coefficient of coumarin in the film, by minimising the sum of 329 the square of the differences between measured and predicted values, using the Levenberg– 330 Marquardt algorithm, and taking D as adjustable parameter. Modeling was performed using 331 Matlab software (The Mathworks, Natick, MA).

332

333 *2.6. Statistical analyses*

334 The data were analyzed using an independent sample t-test with the statistical software SPSS 335 13.0 (SPSS Inc., Chicago, IL). A standard deviation (p-value < 0.05) at the 95% confidence level 336 was used to compare all parameters analysed (water vapour permeability, mechanical properties, 337 surface properties and parameters of the kinetics of release) between irradiated and non-338 irradiated films in the presence and absence of coumarin.

339

- 340 **3. Results and discussion**
- 341 **3.1. How irradiation and coumarin addition influence chemical and structural organisation** 342 **of polymer-blend network**

343 **3.1.1. Free radical generation**

344 Electron spin resonance (ESR) is an appropriate tool to detect and identify the generation of free 345 radicals in a polymer structure. The ESR spectra of chitosan-gelatin blend film incorporated with 346 coumarin before and after irradiation (60kGy) are displayed in **Fig.1**. A very weak ESR signal is 347 detected, at 3365 G, in the case of non-irradiated films. Contrarily, just irradiated film displays a 348 prominent peak at the same position suggesting the presence of free radical species induced by 349 the irradiation treatment. This peak remains visible even after 3 months of storage revealing that 350 free radicals are still present in the films.

351 Recently, we reported that the peak intensity, determined at 3500 G, for irradiated chitosan-352 gelatin film increases with increasing the irradiation dose Benbettaieb et al. (2015a).

353 The analysis of the peak to peak amplitude between film containing or not containing coumarin 354 shows that, after irradiation at 60kGy, the peak amplitude is 2 to times higher for irradiated 355 films compared to non-irradiated ones, for film without coumarin and for film containing 356 coumarin, respectively.

357 The detection of free radicals generated within the films may contribute to clarify the role of 358 these reactive species to initiate the reaction sequence inducing change in the chemical and 359 structural organisation of the biopolymer-blend network. A new arrangement in the structure is 360 expected after irradiation, which could favour the linkage between the biopolymers chains or 361 between the biopolymers chains and the active compound. Even if the water content of film was 362 quite low (about 12%), possible hydroxyl-coumarin obtained after hydroxylation reaction by OH 363 radical produced from water radiolysis could react with polysaccharide and protein network. As

364 coumarin is a phenolic substance made of fused benzene and alpha pyrone ring, their phenolic 365 groups can be easily converted to quinone in presence of peroxide radicals (produced by 366 irradiation), via quinone-mediated reactions. Indeed, ahe slight change in the film colour after 367 irradiation is observed. This could be more probably attributed to quinone generation from 368 antioxidants more than Maillard reaction because the temperature involved in the film making 369 and irradiation process remained lower than 50°C. In addition to providing a source of stable free 370 radicals, quinones are known to complex irreversibly with nucleophilic amino acids in protein. 371 The reaction mechanism involves an initial oxidization of phenolic structures to quinones, which 372 can readily react with nucleophiles from reactive amino acid groups in protein: sulfuhydryl 373 group in cystein, amino group of lysine and arginine, amide group from aspartic and glutamic 374 acids, indole ring of tryptophan and imidazole ring from histidine (Zhang et al., 2010).This tends 375 to be confirmed as our films gain a yellowish colour after irradiation. Sahu et al. (2009) showed 376 the efficiency of microwave irradiation for the oxidation of phenol to quinone after free radicals 377 generation in the presence of hydrogen peroxide. Casimiro et al. (2010) showed that, in acidic 378 medium, NH_3^+ groups (from deacetylated units) of chitosan are able to be involved in some 379 interactions during irradiation. As displayed by Madeleine-Perdrillat et al. (2015), even at low 380 water content, molecular mobility in chitosan films remains quite high and allows the supposed 381 free radical mobility in the film. Knowing free radicals are present after irradiation, they could 382 originate sequences of reactions within and between polymer chains and antioxidant which could 383 in turn affect the structural properties of the final network. These structural and functional 384 properties modifications have been assessed in the followings.

385

386 **3.1.2. Changes in chemical structure by molecular interaction analysis**

387 GPC-MALLS was used to determine the molecular weight parameters of chitosan-gelatin film 388 before after irradiation and in the presence of coumarin. The typical elution profile (15 – 19 mL) 389 of chitosan-gelatin film was monitored by the light scattering (detector 90° degree), refractive 390 index and UV at 280nm detectors (data not shown). The results are tabulated in **Table 1**. The 391 weight average molecular weight for chitosan gelatin film in the presence and absence of 392 coumarin was similar and almost an identical value was obtained following irradiation at 60kGy. 393 The results given in **Table 1** show that there is no significant difference of for the molecular 394 weight, the polydispersity and the z-radius between irradiated (and non-irradiated films are 395 observed for the control). So irradiation induce neither covalent reticulation nor biopolymer 396 degradation under the conditions employed in this study. The incorporation of coumarin has no 397 additional effect on the molecular size of the biopolymers even when coupled with irradiation 398 process

399

400 Fourier transform infrared spectroscopy (FTIR) was used in order to assess the possibility of 401 interactions and the nature of linkage between polysaccharide-protein network and coumarin 402 following irradiation treatment. FTIR spectra of irradiated and non-irradiated chitosan-fish 403 gelatin blend films in the presence and absence of coumarin are displayed in **Fig. 2**. The 404 spectrum of non-irradiated control film showed characteristic peaks: 3300-3360 cm⁻¹, assigned 405 to v_{OH} stretching of free water and v_{NH} stretching of amide A, 2915-2935 cm⁻¹, assigned to v_{CH} 406 asymmetric/symmetric stretching of amide B, 1550-1680 cm⁻¹, assigned to C=C and C=O 407 stretching of primary and secondary amine NH band of amide I, 1550-1610 cm⁻¹, assigned to δ NH of amide II and 1240-1340 cm⁻¹, assigned to aromatic primary amine, CN stretch of amide 109 III (Coates, 2000; Benbettaieb et al., 2015a). The peak observed in 1034 cm⁻¹ is related to 410 possible interactions arising between plasticizer (OH group of glycerol) and polymer structure 411 via hydrogen bonds (Cerqueira et al, 2012). The amide B v_{CH} (2932 cm⁻¹) is slightly shifted to 412 higher wavenumber (2941 cm⁻¹) after incorporation of coumarin. However, no change is 413 observed on amide-I, amide-II and amide-III peak position. Similar results was founded by

414 Benbettaieb et al.(2015b) after incorporation of ferulic acid in chitosan-gelatin film. Contrarily, 415 other author found a shift on amide I band, attributed to interaction between chitosan (amide) 416 and starch or ferulic acid (hydroxyls) in the case of starch-chitosan blend film incorporating 417 ferulic acid (Mathew and Abrahman, 2008); or to caffeic acid oxidation inducing protein 418 crosslinking (Nuthong et al, 2009); or to crosslinking in gelatin gel after UV irradiation (Bahat 419 and Karim, 2009). Therefore, FTIR analysis only shows a weak modification in the structure of 420 the final network without new significant linkage. Only the amide A and amide B (and amide II 421 for control film) groups exhibit a shift after both antioxidant addition and irradiation treatment.

422

423 **3.1.3. Thermo-gravimetric analysis (TGA)**

424 Thermo-gravimetric analysis was performed in order to study the effect of coumarin addition 425 before and after irradiation on the thermal stability of blend chitosan-fish gelatin films. **Fig. 3** 426 shows two main stages of weight loss events for all films. The first stage occurs over a 427 temperature range of around 51-121°C and results in a weight loss (∆w1) of approximately 5.2- 428 7.4%. It is associated with the loss of acetic acid and free water sorbed in the film. These results 429 are in agreement with Inamura et al. (2013), who observed similar behavior for biocomposite 430 films prepared with gelatin and nut wastes as fiber source (46-140 °C). Barreto et al. (2003) and 431 Pena et al. (2010) also showed similar results for gelatin film, from 25 to 200°C. These 432 temperature ranges differences can be attributed to the variation of the initial water content as 433 well as the plasticizer used. The second stage of weight loss (∆w2= 42.3-53.8%) occurs in the 434 temperature range from 215 up to 330°C. This is most likely due to to the degradation of the 435 polysaccharide and protein backbones as well as the evaporation and thermal degradation of 436 glycerol (from 177-211^oC up to 450^oC (Maturana and Pagliuso, 2011)) and also structurally 437 bound water evaporation. Pure coumarin (powder) exhibits a single stage in weight loss, with 438 decomposition starting at 160-180°C and finishing at about 230°C **(Fig. 3)**. Only weak

439 difference was observed regarding thermal decomposition temperatures and weight loss (∆w1 440 and ∆w2) when control film is compared to the films with coumarin due to the low coumarin 441 content (47mg/g). Opposite results were observed in the case of skin gelatin after addition of star 442 anise extracts (Hoque et al., 2011) and green tea extract (Wu et al., 2013). They suggested that 443 interactions occurring between phenolic compounds and gelatin yielded to a stronger film 444 network and therefore a higher heat resistance of the films. The above studies are comparable to 445 those we recently reported on the same gelatin- chitosan film which showed improved thermal 446 stability following the addition of quercetin (Benbettaieb et al., 2015b). After irradiation, thermal 447 degradation temperature of the control film is improved, associated with a decrease observed on 448 weight loss (∆w1 and ∆w2). This result suggests the apparition of new bonds, thermally more 449 resistant to heat than initial bonds existing before irradiation which enhances the thermal 450 properties. Similar result was found by Inamura et al. (2013) in the case of composite gelatin-nut 451 shell fiber after 40 kGy irradiation dose. Inversely, we cannot observe any significant 452 modification of thermal stability for film containing coumarin after irradiation. Benbettaieb et al. 453 (2015b) showed a reverse tendency for chitosan-gelatin film containing ferulic acid after 60kGy 454 irradiation dose. Finally, from the above structural and thermal analysis, we can conclude that 455 the interaction between coumarin and polymer chains is very weak and no covalent or strong 456 linkage occurred following irradiation. For this reason, a complementary analysis must be 457 undertaken to better understand the effect of both irradiation and coumarin addition on functional 458 properties of films.

459

460 **3.2. Impact of both irradiation and coumarin addition on functional film properties**

461 **3.2.1. Surface properties and wettability**

462 The contact angle (θ) value obtained after deposition of a water drop on the film surface 463 indicates the surface hydrophobicity. To estimate the resistance of films to liquid water, the

464 swelling index and swelling rate were also determined from the droplet volume kinetics, along 465 with the contact angle values at the initial time of deposit (0 s) and at a considered metastable 466 equilibrium (20 s). Results for all films are summarized in **Fig. 4**. For untreated films, the contact 467 angles (at 0 and 20s) significantly decrease (p<0.05) after incorporation of coumarin. 468 Furthermore, no swelling is observed for the control film. Swelling index and swelling rate 469 significantly (p<0.05) increase to $52\pm4\%$ and to $68\pm24x10^{-3}$ μ L/s, respectively after addition of 470 coumarin. After irradiation, the contact angles (at 0 and 20s) significantly decrease for all films 471 (decreases is not significant only for the contact angle at 20s for coumarin film). Whereas, 472 swelling index and swelling rate tend to increase for all films, but they are only significant 473 (p<0.05) for the control films. To better understand the effect of coumarin on film surface 474 properties under electron beam irradiation, the surface tension was also determined. Surface 475 tension does not show significant modification after incorporation of coumarin. However, a 476 slight increase is observed in the polar component. The presence of this antioxidant seems to 477 slightly contribute to the hydrophilicity of the film. Similar behaviour was recently reported by 478 Benbettaieb et al.(2015b) in our study on the same film but in the presence of ferulic acid. 479 Irradiation induces a decrease of the contact angle value with water, concomitant to an increase 480 in the polar component of the surface tension for all films. It can be attributed to a reorientation 481 of polar groups at the film surface, hence increasing the polar component of the surface tension. 482 Thus irradiation increases wettability of the films.

- 483
- 484 **3.2.2. Water vapour permeability**

485 **Table 2** displays the WVP of non-irradiated and irradiated films in the presence and absence of 486 coumarin for the two RH gradients studied (0-30 % and 30-84 %). For the 0-30%RH gradient, the 487 WVP of non-irradiated film containing coumarin $(0.47\pm0.03 \times 10^{-11} \text{ g.m}^{-1} \text{ s}^{-1} \text{ Pa}^{-1})$ is in the same 488 range to that of control film $(0.52\pm0.1x10^{-11} \text{ g.m}^{-1} \text{ s}^{-1} \text{.} \text{Pa}^{-1})$. Inversely, Wu et al. (2013) noticed a

489 decrease of 16 % in the WVP for films composed of silver carp (*Hypophthalmichthys molitrix*) 490 skin gelatin containing green tea extract (0.7%). Other authors did not observed any change in 491 the water vapour permeability when ferulic acid was added to to gelatin films (Cao et al. 2007), 492 to soy protein films (Ou et al. (2005), or to caseinate based films (Fabra et al., 2011). After 493 irradiation we did not observed any modification in WVP (0-30% RH gradient) for control film 494 and for film containing coumarin. Furthermore, the 0-30% RH gradient correspond to water 495 activity average equal to 0.15, which is in the BET domain (water contained in the film is only 496 involved in the structure organisation and not available for reaction), thus we propose that only 497 water involved in structure with weak plasticization of the network by water. In this domain, 498 neither irradiation nor coumarin addition affects the water barrier properties of the films. On the 499 other hand, higher RH gradient of 30-84% induces a rise of the WVP of all films (with or 500 without coumarin addition) after irradiation treatment. WVP increases from 2.41±0.44 to 501 23.06 \pm 0.85 x10⁻¹¹ g.m⁻¹.s⁻¹.Pa⁻¹ and from 2.23 \pm 0.65 to 24.8 \pm 2.11 x10⁻¹¹ g.m⁻¹.s⁻¹.Pa⁻¹, 502 respectively for control film and film with coumarin after 60kGy irradiation dose. Due to the 503 effect of irradiation, the barrier properties are mainly related to the increasing the water content 504 that induces the plasticization of the film during permeation. The 30-84% RH gradient 505 corresponds to a mean water activity average equal to 0.57, which is in the plasticization domain 506 of the network by water. But the effect of coumarin addition in the network on the transfer 507 phenomena is not significant.

508

509 **3.2.3. Mechanical properties**

510 Mechanical parameters (tensile strength (TS), Young's Modulus (YM) and elongation at break 511 (E)) of all studied films are given in **Table 2**. TS, YM and E of control film (non-irradiated, 512 without coumarin) were 25.9 ± 3.9 MPa, 1523 ± 266 MPa and $2.2\pm0.4\%$, respectively. Jridi et al. 513 (2014) found similar value of %E (2.7 \pm 0.5%) but higher value of TS (44.3 \pm 1.2 MPa) for

514 chitosan-skin fish gelatin blend film (50:50 w/w) that could be due to film thickness or molar 515 mass of polymer. Compared to the control film, no significant modification is observed on the 516 mechanical parameters (TS, YM and %E) after coumarin addition. An opposite tendency was 517 observed by Benbettaieb et al. (2015b) who showed that TS increases significantly when ferulic 518 acid was added in to the chitosan-gelatin film. Only the TS of control film increases 519 significantly with the increasing dose at a 60kGy. When irradiation doses (40 and 60kGy) were 520 applied on film containing coumarin, no significant ($p<0.05$) modification of mechanical 521 parameters was observed. This increase of film stiffness and resistance is in accordance with the 522 improvement of thermal stability of control film after 60kGy irradiation dose, previously 523 observed from TGA analysis.

524 Finally, results from functional and structural properties are still less consistent to make any 525 hypothesis related to the crosslinking reaction between polymers chains and coumarin under 526 irradiation. Only few interactions occur (probably modified by hydrogen bonds) between the 527 different reactive compounds and some orientation of polar groups to the surface, which enhance 528 the film wettability and hydrophilicity. The interactions between polymer and coumarin after 529 irradiation, even if they seem to be weak could nevertheless affect the release of the antioxidant 530 into aqueous medium.

531

532 **3.3. Influence of irradiation on the coumarin release in aqueous media**

533 The release experiments were performed three months after film irradiation. All data are 534 summarized in **Table 3.** The amount of coumarin in the non-irradiated film $(10.1 \pm 1.5 \text{ mg/g of})$ 535 film) determined after complete solubilisation of films in acetic acid solution (experimental 536 value) was significantly ($p<0.05$) lower than the theoretical value (47.1 \pm 4.7 mg/g of film) 537 calculated from the film formulation. This means that, about 80% of coumarin disappeared, 538 probably due to oxidation during storage (**Table 3)**. After irradiation, measured (real) amount of

539 coumarin in the films is far less reduced, as only 35% is lost. Thus, irradiation protects coumarin 540 against degradation during the time of storage (3 months). It could be that coumarin make some 541 interaction with polymers and/or with free radical and thus is less available to be oxidized during 542 storage period. This result suggests that irradiation may act as a safeguard method of antioxidant 543 when this later is encapsulated in hydrocolloid films. This can be considered as a good way to 544 protect active compound and to ensure its quality until final consumer. As the difference 545 observed between theoretical and experimental content of coumarin in the film, only the 546 experimental concentration is considered for the study of the release. The release kinetics of 547 coumarin from chitosan-gelatin based films (non-irradiated and irradiated at 40 and 60kGy) in 548 water medium (at pH=7) are displayed in **Fig.5.** Release kinetics of coumarin exhibited the 549 typical shape of non-time-dependent and non-concentration-dependent diffusion. The content of 550 coumarin remaining in the film after release significantly ($p \le 0.05$) increased from 1.7 \pm 0.6 to 551 7.17 \pm 0.5 and to 12.6 \pm 1.7 mg/g of film, respectively after 40 and 60kGy irradiation doses. This 552 could be explained by the interaction between the polymer chains and this antioxidant, favoured 553 by the irradiation process and because the initial concentration is higher. This could therefore 554 modify the film structure organisation and the release mechanisms of antioxidant from the film. 555 Furthermore, the effective diffusion coefficients (D) of the coumarin in the films were calculated 556 from the release kinetics by fitting experimental data using Eq.6 and are given in **Table 3**. The 557 diffusivity is related to the molecular mobility within the polymeric network and could be related 558 to several factors such as molecular weight, structural characteristics of the matrix and solubility 559 of this antioxidant. We considered here that the partition coefficient did not affect the transfer as 560 the concentrations involved are always much lower that the solubility limit of the coumarin in 561 water media. As it can be observed, the effective diffusion coefficient of coumarin significantly 562 decreased (p<0.05) from 3.26 \pm 0.74 to 1.87 \pm 0.48 and to 2.04 \pm 0.05x10⁻¹¹m².s⁻¹ respectively after 563 40 and 60kGy irradiation doses. This is also in agreement with the increase of the coumarin

564 content remaining in the film after the release (by chemical or physical entrapment).This 565 decrease of diffusion coefficient could be related to the irradiation treatment which limits the 566 mobility of coumarin and therefore decreases the apparent diffusivity. Tin Wui et al. (2002), 567 worked on the influence of microwave irradiation on the drug release properties of 568 polysaccharide beads and showed that the release-retarding property of alginate and alginate– 569 chitosan beads was significantly enhanced by subjecting the beads to microwave irradiation. 570 They showed that microwave technology can be employed in the design of solid dosage forms 571 for controlled-release application without the use of noxious chemical agents. In our case, 572 irradiation could favor the interaction between coumarin and biopolymer via free radical 573 mediated mechanism. Hence, coumarin is more linked and consequently, more protected and less 574 mobile. The effect of irradiation also modified the surface properties by increasing its polarity 575 and then swelling phenomenon occurs too quickly to affect the diffusion determination.

576 Despite the swelling, the film remains intact and no dissolution or network structure destruction 577 was observed during the kinetic of release. In non-irradiated film, as the structure is less dense, 578 water can easily enter into the network and favour the polymeric chain mobility and thus the 579 coumarin diffusion through the hydrated films is greater. Irradiation allowed to delay by 50% the 580 release time. So, film irradiation after optimization, would be an effective process for controlled 581 release of active naturals antioxidants in aqueous foods or even for medical applications.

582

583 **4. Conclusions**

584 Chitosan and fish gelatin films encapsulating coumarin were prepared as an active biobased film. 585 After film drying, irradiation using electron beam was applied at 40 and 60kGy. This work 586 aimed to investigate the coupled effect of irradiation and of the presence of the active compound 587 on the structure and functional properties of the films. Electron Spin Resonance (ESR) displayed 588 the free radical formation during irradiation in films. Coumarin did not affect the thermal 589 stability of films whereas irradiation slightly improved it. Both addition of coumarin and

590 irradiation decreased the contact angle with water and increase the polar component of the 591 surface tension of films, as well as the swelling index and rate. This is attributed to a 592 reorientation of polar groups at the film surface. From water barrier analysis, neither irradiation 593 nor coumarin addition affected the water vapour permeability at low RH gradient. However, a 594 higher RH gradient (30-84%) induced a rise of the WVP of all the films after irradiation 595 treatment that is mostly related to the surface properties and film wettability. Incorporation of 596 coumarin did not affect the mechanical properties of films on the contrary to irradiation, but very 597 weakly. The interactions between biopolymers and coumarin after irradiation affected the release 598 of the antioxidant into the aqueous medium. The content of coumarin remaining in the film at 599 equilibrium after release significantly increased when film were irradiated, from 17% to 32% 600 mg/g of film, inversely, the effective diffusion coefficient of coumarin decreased by 1.6 times. 601 Irradiation, also displayed that it is an efficient process to prevent coumarin degradation during 602 the storage of films, as more than 60% of the antioxidant was preserved compared to non-603 irradiated films.

604

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612

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2 **Figure captions**

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20 **Tables captions**

21 **Table 1:** Number and weight average molar mass (Mn, Mw), polydispersity (Mw/Mn) and z-22 average mean square radius (Rz) for irradiated and non-irradiated control films and for irradiated 23 coumarin films, determined from SEC-MALLS analysis.

24 **Table 2:** Thickness, water vapour permeability (WVP) and mechanical properties (Tensile 25 strength (TS), Young's Modulus (YM) and elongation at break (% E)) of irradiated and non-

33

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2 **Figure 1**

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Absorbance

 \overline{c}

 $\begin{array}{c} 8 \\ 9 \end{array}$ 10

15

1 **Table 1**

2 Mean (relative error%)

4 **Table 2**

6 Values are given as mean \pm standard deviation. Means with the same Arabic letter in the same column are not significantly different at p<0.05.

8 **Table 3**

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